# Origin of High Melting Point of Substituted Poly(lactic acid) with Isopropyl Side Chains Revealed by Synchrotron WAXD/SAXS Simultaneous Measurements

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# 1 Introduction

Recently, the side-chain-substituted poly(lactic acid)s (SPLAs) [1,2] have attracted attention as novel bioplastics to overcome weakness of poly(L-lactic acid) (PLLA), a representative bioplastic. However, their possibility as crystalline plastics still remains unexplored. We reported that the crystalline SPLA with isopropyl side chains [P(L-2H3MB)] had much higher melting temperature  $(T_m)$  and crystallization rate as compared to PLLA, from which a great potential as high-performance bioplastics was demonstrated [1]. In this study, we investigated the nonisothermal crystallization process of P(L-2H3MB) by WAXD/SAXS time-resolved (TR) simultaneous measurements. We discuss the origin of high  $T_{\rm m}$  of P(L-2H3MB) based on TR-WAXD/SAXS and thermal analysis.

## 2 Experiment

P(L-2H3MB) with  $M_n = 1.4 \times 10^5$  g/mol and  $M_w/M_n = 1.4$  was synthesized from L-valine [1].

TR-WAXD/SAXS simultaneous measurements were performed at KEK PF BL-6A ( $\lambda = 0.1500$  nm) using an FP84HT TA Microscopy Cell (METTLER TOLEDO) for cooling from 240 to 30 °C at 5 °C/min, during which data acquisition and interval times were set to 5 and 1 s, respectively. The temperature of samples packed in a washer was monitored by a resistance temperature detector.

Differential scanning calorimetry (DSC) measurements were conducted in lab using a power-compensated DSC (Diamond DSC, PerkinElmer Inc.) in the non-isothermal condition with the same temperature range (240 to 30 °C) and cooling rate (5 °C/min) as TR-WAXD/SAXS.

#### 3 <u>Results and Discussion</u>

Fig. 1 shows TR-WAXD profiles during the cooling process from the melt for P(L-2H3MB). The crystallization of P(L-2H3MB) started (appearance of diffraction peaks) around 185 °C and the degree of crystallinity ( $X_c$ ) increased (increases in peak areas) with decreasing temperature. The crystal modification ( $\beta$ -form) was not changed during the non-isothermal crystallization. Note that shifts of peaks to higher  $2\theta$  (shorter spacing) with decreasing temperature would be attributed to thermal shrinkage of samples. The temperature dependence of  $X_c$ was obtained from TR-WAXD profiles. The peak temperature in  $dX_c/dT$  is 180 °C, which agrees with that in the DSC cooling curve. The equilibrium crystallization enthalpy ( $\Delta H_c^\circ$ ) was evaluated from the linear relationship between  $X_c$  and the corresponding crystallization enthalpy  $(\Delta H_c)$ . The obtained  $\Delta H_c^{\circ}$  was converted to the equilibrium melting enthalpy  $(\Delta H_m^{\circ})$  considering a difference between specific heats in the glassy and rubbery states.

The equilibrium melting temperature  $(T_{\rm m}^{\circ})$  was evaluated from the Gibbs–Thomson  $(T_{\rm m} \text{ vs } l_{\rm c}^{-1})$ , where  $l_{\rm c}$  is the lamella thickness obtained form SAXS) and Hoffman– Weeks plots  $(T_{\rm m} \text{ vs } T_{\rm c})$ . The equilibrium melting entropy  $(\Delta S_{\rm m}^{\circ})$  was obtained from the relationship  $T_{\rm m}^{\circ} =$  $\Delta H_{\rm m}^{\circ}/\Delta S_{\rm m}^{\circ}$ . As a result, a smaller  $\Delta S_{\rm m}^{\circ}$  of P(L-2H3MB) than PLLA was found to give a higher  $T_{\rm m}$  of P(L-2H3MB).



Fig. 1: TR-WAXD profiles during cooling process from the melt for P(L-2H3MB).

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