

Detecting modulation of orbital magnetic moments and magnetic anisotropy in Co₂FeSi /PMN-PT via *Operando* X-ray magnetic circular dichroism

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Combining ferromagnetic and ferroelectric properties has been investigated as multiferroics, which provides considerable advantages of magnetization control by an electric field (E) as a low-energy power consumption operation without electric current. As spintronics research pursues the suppression of power consumption for device operations, the manipulation of the spins by E is one of the solutions for future device applications. Recent developments have focused on the modulation of magnetization by a low electric field. Recently, a study reported that Co₂FeSi / Pb(Mg_{1/3}Nb_{2/3})O₃-PbTiO₃ (PMN-PT) interfacial multiferroics exhibited a giant magneto-electric (ME) effect of 1.8×10^{-5} s/m [1], where Co₂FeSi (CFS) is a Co-based Heusler alloy that was expected to have high spin polarization at room temperature and high Curie temperature. The origin of the large ME effect in CFS-based interfacial multiferroics was associated with strain-mediated magnetic anisotropy modulation. However, the microscopic origin of the giant ME effects in the CFS/PMN-PT interfacial multiferroics is still unknown. To initiate novel research into the physics of the relationship between the lattice distortion and orbital magnetic moments (m_{orb}), the anisotropic m_{orb} needs to be explored even in interfacial multiferroics [2]. Since *operando* X-ray magnetic circular dichroism (XMCD) can probe element-specific orbital modulation when a reversible strain is applied, we developed the XMCD system with applying E .

X-ray absorption spectroscopy (XAS) and XMCD measurements of the Fe and Co L -edges were performed using the KEK-PF BL-7A beamline, Japan, at room temperature. A magnetic field was applied along the incident polarized soft X-rays along the normal direction of the surface of the sample. The electrodes were mounted on the surface of the sample and the rear of the substrate to perform the E -induced XMCD measurements [3].

Figure 1 shows the Fe and Co L -edge XAS and XMCD spectra obtained in the partial fluorescence yield mode under an applied electric field E of ± 0.8 MV/m. The sample surfaces were connected to the ground, and E was applied to the back side of the PMN-PT substrates. The XAS spectral line shapes originated from metallic Fe and Co features in CFS. The spectral line shapes of XMCD and the slight change in the L_3 -edge peak were modulated by E only at the Fe L -edge, despite the fixed sample measurement position. These results indicated that the m_{orb} were modulated by the applied E , thereby resulting in changes in magnetic anisotropy. The modulation of m_{orb} by $0.01 \mu_B$ under an applied E was related to the induced lattice distortion from the PMN-PT substrates. In contrast

to the case of Fe, no spectral changes were detected for the Co L -edge XMCD [4].

The element-specific XMCD hysteresis curves at the Fe and Co L_3 -edges during the application of E . In the case of $+0.8$ MV/m, easy axis behavior was clearly observed, whereas the -0.8 MV/m case indicated the hard axis. This result showed reproducible behavior of the changes in the magnetic easy-axis direction. In the case of Fe shown in Fig. 1c, the vertical axis strength was slightly different between ± 0.8 MV/m, thus affecting the changes in XMCD in the Fe L_3 -edge intensity, whereas the hysteresis curves of Co by ± 0.8 MV/m overlapped above the saturation magnetic fields (Fig. 2d). Therefore, only the change in m_{orb} for Fe was experimentally found [4].

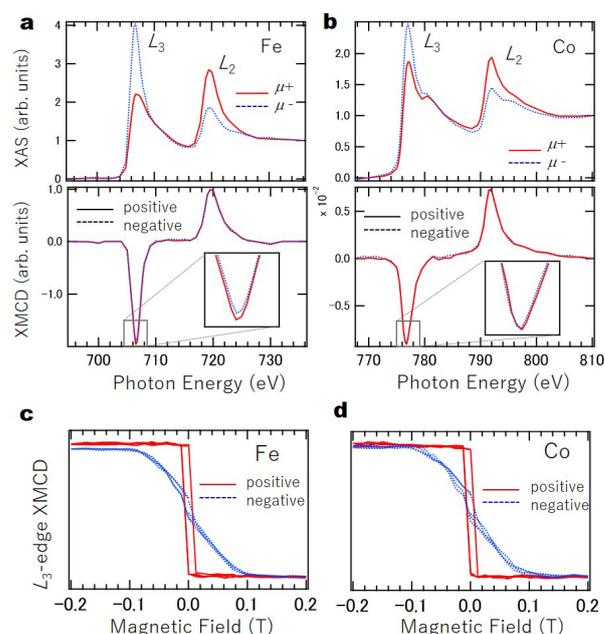


Fig. 1. XAS and XMCD under an applied electric field in Co₂FeSi/PMN-PT. (a) Fe and (b) Co L -edges at ± 0.8 MV/m. Expanded views around L_3 edges are shown in the bottom panels. Magnetic field dependence of L_3 -edge XMCD under an applied electric field. (c) Fe and (d) Co L -edges at ± 0.8 MV/m [4].

References

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