Operando XMCD for magnetic anisotropy control by reversible strain at Fe₃Si/PMN-PT interface

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Artificial ferromagnetic/ferroelectric multiferroic heterostructures have been widely investigated from both fundamental and technological aspects. The electric field (E)control of magnetism through the interfacial strain modulates the magnetic anisotropy in the magnetic layer. Controlling the magnetic anisotropy by the interfacial strain related to the orbital (m_{orb}) or quadrupole (Q) moments has a potential for future devices using both spins and orbitals. The magnetic anisotropy (K) can be formulated by orbital moment anisotropy Δm_{orb} and Q,

$$K = \frac{1}{4} \xi \Delta m_{\rm orb} + \frac{\xi^2}{\Lambda E} Q \langle S \rangle,$$

using spin-orbit coupling ξ , exchange splitting ΔE and spin $\langle S \rangle$ [1]. The relationship between strain and m_{orb} or Q has not been clarified because there were few tools to probe the changes of m_{orb} [2]. We have developed the *E*-induced x-ray magnetic circular dichroism (EXMCD) technique in order to apply *E* to ferroelectric substrate Pb(Mg_{1/3}Nb_{2/3})O₃-PbTiO₃ (PMN-PT), which tunes the interfacial lattice constants of the Fe₃Si magnetic layer, which can be recognized as Fe substitution into Heusler alloy Co₂FeSi [3]. In this study, we discuss the microscopic origin of inverse magneto-striction effects or orbital-elastic effects concerning the m_{orb} and quadrupole moments by using the EXMCD.

We prepared the samples of 30-nm-thick Fe₃Si (422) layer grown on single-crystal PMN-PT(011) substrates with the insertion of 0.3-nm-thick Fe layer by molecular beam epitaxy. The *E*-induced modulation of the in-plane magnetic properties was characterized by XMCD, where the XMCD measurement was performed at BL-7A in the Photon Factory (KEK). The partial-fluorescence-yield mode was adopted to probe the signals more than 10 nm below the sample surfaces. To apply an *E* to the PMN-PT substrate along the [011] direction, a Au(100 nm)/Ti(3 nm) electrode was deposited on the backside of the PMN-PT substrate, where the Fe₃Si film was utilized as a top electrode [3,4].

The magnetization measurements revealed that the magnetic easy axis is along the PMN-PT $[01\overline{1}]$ in-plane direction at E = 0 kV/cm. By applying E of -8 kV/cm, the easy axis changes 90° along [100] direction within the inplane through the changes of lattice strain of 0.1 % order. The Fe L_3 -edge XMCD hysteresis curves also trace the same changes of magnetic anisotropy of 1.6×10^3 J/m³. Interestingly, the electric field dependence of easy axis direction is opposite to the case of Co₂FeSi/PMN-PT [3]. Furthermore, the XMCD line shapes remain unchanged by applying E, which corresponds to the changes originating not from m_{orb} but from the modulation of charge distribution forming quadrupole moments. Since the electronic structures in Fe₃Si deduced

from the band-structure calculation are quite different from those in Co₂FeSi, m_{orb} in Fe₃Si is not scaled in linear relationship by strain unlike the case of Co₂FeSi [3]. These facts are affected to the *E* dependence in the XMCD line shapes. Therefore, applying reversible strain by *E* into Fe₃Si does not contribute to the orbital striction, which might be originated from the quadrupole modulation from two kinds of Fe site facing on Si atoms.



Fig. 1, XAS (a) and XMCD (b) under an applied electric field in Fe₃Si/PMN-PT. (c) Magnetic-field dependence of Fe L_3 -edge XMCD under an applied E of \pm 8 kV/cm with reversibly switching the applying electric field.

References

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