# The Chemical Environment about Cd atoms in Cd Chemical Bath Treated CuInSe<sub>2</sub>

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### **Introduction**

CdS buffer layers located between the p-type Cu(In;Ga)Se<sub>2</sub> (CIGS) absorber layer and the n-type ZnO window layer have become an integral part of high-efficiency polycrystalline CIGS solar cells. Experimentally it has been found that the highest conversion efficiencies occur in cells with chemical-bath deposited (CBD) CdS buffers. Even though the amount of Cd incorporated into the cells is minute, an effort has been made to understand the reasons for the efficacy of the CDB-CdS buffer as a preamble to find alternative, less toxic substitutes. Several recent reports have concluded that treatment by (partial electrolyte) PE-Cd alone, or in combination with conventional CBD-CdS deposition results in improvement in either cell performance or reproducibility[1]. An outstanding question that must be answered before detailed investigation of the underlying mechanism can begin is where in the CIGS matrix the CBD deposited Cd resides. We have used near-edge x-ray absorption fine structure (XANES) at the Cd-L<sub>3</sub> edge (3537 eV) to selectively investigate the local atomic environment about Cd atoms in CBD-Cd treated polycrystalline CuInSe2 and CuGaSe2 deposited on Mo-coated soda-lime glass substrates which were then compared with first-principles calculations.

## Experiment

Both the CuInSe<sub>2</sub> and CuGaSe<sub>2</sub> samples used in this study were deposited by the three-stage method onto Mo-coated soda-lime glass substrates.[2] The as-grown samples were placed into a CBD-Cd chemical bath kept at a temperature of  $40^{\circ}$  for 40 minutes. The Cd precursor used was CdI<sub>2</sub> ( $0.7 \times 10^{-3}$  mol/l) in an NH<sub>4</sub>OH 0.5 mol/l solution. After deposition, the samples were then introduced into a vacuum chamber at beam line BL11B at the Photon Factory in Tsukuba, Japan. The Cd-L<sub>3</sub> edge (3537 eV) was excited in glancing incidence mode and fluorescence data was taken using a windowless Ge SSD.

### **Results and Discussion**

Fig. 1 show both experimental data for the Cd-L<sub>3</sub> edge taken for both CuInSe<sub>2</sub> and CuGaSe<sub>2</sub> CBD-Cd treated films in fluorescence mode. In addition to the experimental data the results of theoretical calculations done using the *ab initio* code feff 8.10[3] are shown for Cd residing in different places in the host matrix as well as Cd in

metallic and oxide form. The feff calculations were performed using a self-consistent field approach as well as full multiple scattering on a cluster of radius 0.8 nm.

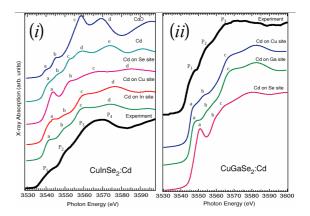


Figure 1: Experimental and simulated Cd-L<sub>3</sub> XANES spectra for CBD-Cd treated CuInSe<sub>2</sub> (i) and CuGaSe<sub>2</sub> (ii) films

#### **Conclusion**

We have measured the local structural environment of Cd in CBD-Cd treated CuInSe<sub>2</sub> and CuGaSe<sub>2</sub> polycrystalline samples using XANES at the Cd-L<sub>3</sub> edge and compared the experimental results to theoretical simulations. Similar results were found for both materials, namely that Cd most likely resides on the cation sites alone. The current experiments cannot distinguish between which of two cation sites the Cd ions reside upon. The presence of Cd on both cation sites also implies the near surface layer may be strongly electrically compensated.

# References

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