

Mechanism of ohmic-contact formation between Ti electrodes and $\text{Al}_x\text{Ga}_{1-x}\text{N}$

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1. Introduction

Ohmic contacts between metal electrodes and nitride semiconductors with low resistivities are required to improve the device performance. For *n*-type GaN, Ti-based contacts have been extensively used due to several advantages such as a low work function and cost performance [1-3]. The most important one is that Ti-based contacts decrease the resistance by a chemical reaction and behave as an ohmic characteristic by the annealing above 500 °C [4]. This behavior is explained by the creation of nitrogen vacancies at an interfacial GaN layer by the formation of metallic TiN compounds through the annealing [4,5]. Ti-based electrodes have been also examined for $\text{Al}_x\text{Ga}_{1-x}\text{N}$ with low Al ratio, and the annealing process has been also performed after the deposition of electrodes [6]. To clarify the mechanism of low-resistance ohmic-contact formation at Ti/ $\text{Al}_x\text{Ga}_x\text{N}$ interface, the investigations for energy-band structures are required. In this study, we have investigated the chemical reaction at a Ti/*n*- $\text{Al}_{0.2}\text{Ga}_{0.8}\text{N}$ interface by photoemission spectroscopy to discuss the mechanism of the ohmic-contact formation comparing to the results of the Ti/GaN interface.

2. Experimental

$\text{Al}_{0.2}\text{Ga}_{0.8}\text{N}$ and GaN samples used in this study were grown by metalorganic chemical vapor deposition on *c*-plain sapphire substrates. The thicknesses of both *n*-type GaN and $\text{Al}_{0.2}\text{Ga}_{0.8}\text{N}$ layer were 0.2 μm. To remove the surface contamination, samples were dipped into a boiled 39% HCl solution for 50 min. After that, the 0.3-nm thick Ti layer was evaporated in ultrahigh vacuum (UHV). Photoemission spectroscopy was performed at the beam line BL-1C of Photon Factory in KEK.

3. Results and discussion

Figure 1 shows current-voltage characteristics of Ti electrodes/*n*- $\text{Al}_{0.2}\text{Ga}_{0.8}\text{N}$ before and after the annealing at 500 °C for 10 min. The annealing drastically decreased the resistances, which was consistent with the case of Ti/GaN. The degree of decrease in the resistance of Ti/ $\text{Al}_{0.2}\text{Ga}_{0.8}\text{N}$ was smaller than that of Ti/GaN.

Valence-band spectra for $\text{Al}_{0.2}\text{Ga}_{0.8}\text{N}$ and GaN taken at $h\nu=100$ eV are shown in Fig. 2. Spectral line shapes changed by annealing after the Ti deposition, suggesting the interfacial reaction. Satellite structures in Ti 2*p* photoemission spectra at 500 °C indicate the formation of the TiN compound at the interface. Similar phenomena were observed for Ti/GaN interface [4]. Furthermore, Ga 3*d*, N 1*s*, and Al 2*p* peaks shifted to the lower binding energy by annealing. The amounts of shifts for Ti/ $\text{Al}_{0.2}\text{Ga}_{0.8}\text{N}$ were smaller than those for Ti/GaN. These results indicate that the annealing process

promotes the formation of TiN and AlGaN layers with N vacancies at the interface, resulting in decrease in contact resistivity.

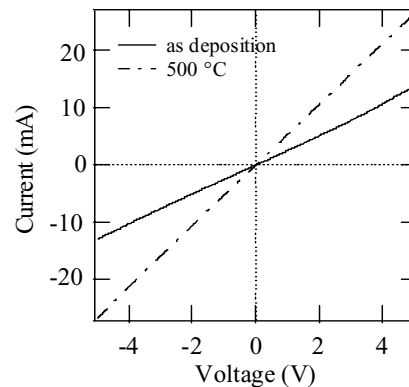


FIG. 1: Current-voltage characteristics of Ti electrodes/*n*-type $\text{Al}_{0.2}\text{Ga}_{0.8}\text{N}$ before and after the annealing at 500 °C

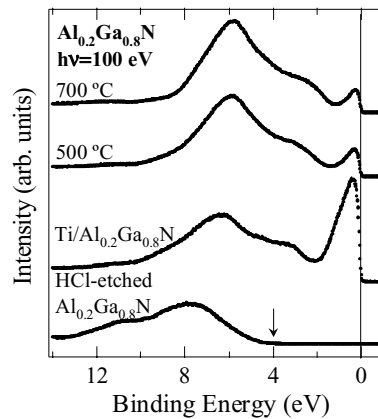


FIG. 2: Annealing-temperature dependence in valence-band spectra. Arrow shows a valence-band maximum for *n*-type $\text{Al}_{0.2}\text{Ga}_{0.8}\text{N}$

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