

## X-ray Magnetic Circular Dichroism study of MnGeP<sub>2</sub> thin film

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### Introduction

Recently the discovery of diluted magnetic semiconductors (DMSs) which show room-temperature ferromagnetism has been a hot issue. These materials are promising candidates for spintronics devices. Some of Mn-doped II-IV-V<sub>2</sub> (chalcopyrite) type DMSs show room-temperature ferromagnetism, and attract much attention. For instance, it is reported that  $T_c$ 's of ZnGeP<sub>2</sub>:Mn and CdGeP<sub>2</sub>:Mn are ~350 K and ~320 K respectively, well above room temperature [1, 2]. It is found that Mn atoms substitute mainly for the II (Zn, Cd) sites in these materials. By *ab initio* calculations based on local density approximation, the ferromagnetism derives from the carriers which arise from the system with vacancies (II, Vc, Mn)GeP<sub>2</sub> or non-stoichiometric (II, Ge, Mn)GeP<sub>2</sub> [3]. In ZnGeP<sub>2</sub>:Mn, 100% Mn substitution i.e. MnGeP<sub>2</sub> was achieved [4].

In this work we have measured soft X-ray absorption (XAS), and soft X-ray magnetic circular dichroism (XMCD) spectra of MnGeP<sub>2</sub> thin film to clarify the electronic state and the origin of ferromagnetism of the MnGeP<sub>2</sub> thin film by studying the magnetic state of the Mn atoms.

### Experimental

The MnGeP<sub>2</sub> thin film was fabricated by the MBE method. The thin film was deposited on a Ge buffer layer at 435 °C which was grown on a GaAs(001) substrate at 380 °C. The Ge buffer layer enables a two-dimensional growth of MnGeP<sub>2</sub> thin film [5]. In order to eliminate surface effects, the sample was capped with a 3nm Ge layer over the thin film. The fabricated MnGeP<sub>2</sub> thin film was transferred into a superconducting magnet under an ultra high vacuum of 10<sup>-9</sup> Torr. The XAS and XMCD spectra were taken in a total electron yield mode at 200 K and 30 K with applied magnetic field at 5.0 T.

We have done a magnetization measurement for the MnGeP<sub>2</sub> thin film in advance. The sample showed  $T_c$  ~320 K and thus it was confirmed to be a room-temperature ferromagnet. Additionally the  $M$ - $T$  curve indicated that there was a component which obeyed the Curie-Weiss law in the sample. This fact suggested that a paramagnetic component existed in the sample.

### Results and Discussion

Figure 1 shows Mn  $L_{2,3}$  XMCD spectra taken at  $T=200$  K and 30 K at  $H=5.0$  T. The intensity has been normalized to at the Mn  $L_3$  edge. The XMCD signal was observed even at 30 K, indicating that the MnP phase is not a dominant component of the sample, because in MnP a ferromagnetic-to-anti-ferromagnetic transition occurs at 47 K. The XMCD spectra have two negative fine structures at the Mn  $L_3$ . Structure A is dominant at 200 K and structure B is dominant at 30 K. This difference may be caused by a paramagnetic component in the sample. Furthermore by applying the XMCD sum rules,  $M_{orb}$  and  $M_{spin}$  were obtained. Assuming that  $n_d \doteq 5.0$ ,  $M_{orb}(200\text{ K}) \sim 0.039 \mu_B/\text{Mn}$  and  $M_{orb}(30\text{ K}) \sim 0.159 \mu_B/\text{Mn}$ . It appears that at low temperature contributions from the paramagnetic component becomes large and that the sizable orbital magnetic moment appears.

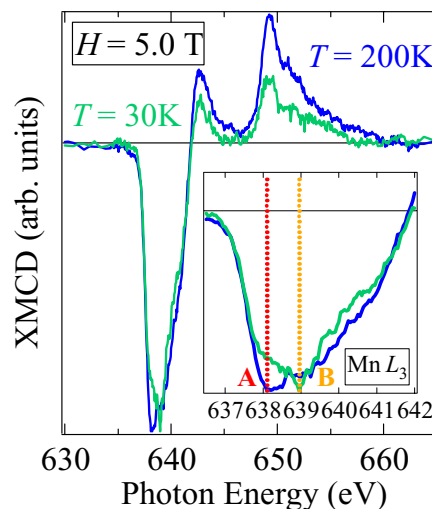


Fig. 1: Mn  $L_{2,3}$  XMCD spectra of the MnGeP<sub>2</sub> thin film

### References

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