# Termination dependence of the Schottky barrier height for La<sub>0.6</sub>Sr<sub>0.4</sub>MnO<sub>3</sub>/Nb:SrTiO<sub>3</sub> heterojunctions

Makoto MINOHARA<sup>1</sup>, Ryutaro YASUHARA<sup>1</sup>, Hiroshi KUMIGASHIRA<sup>\*1-3</sup> and Masaharu OSHIMA<sup>1-3</sup> <sup>1</sup>Department of Applied Chemistry, The University of Tokyo, Tokyo 113-8656, Japan <sup>2</sup>Core Research for Evolutional Science and Technology (CREST), Japan Science and Technology Agency, Tokyo 113-8656, Japan <sup>3</sup>Synchrotron Radiation Research Organization, The University of Tokyo, Tokyo, 113-8656, Japan

## **Introduction**

The height of the Schottky barrier ( $\Phi_{\rm B}$ ) that forms at a metal/insulator junction is an essential and fundamental parameter that dominates the device performance. A perovskite oxide heterojunction (ABO<sub>3</sub>/A'B'O<sub>3</sub>) has two types of interfacial structures that are AO/BO<sub>2</sub>//A'O/B'O<sub>2</sub> and BO2/AO//B'O2/A'O layer sequences, and consequently different interfacial electronic structures emerge depending on the interfacial termination [1]. Thus, the precise determination of the band diagrams for interfacial temination-layer controlled oxide heterojunctions is indispensable for designing spintronic devices, such as tunneling magnetoresistance devices using a half-metallic ferromagnetic material like  $La_{0.6}Sr_{0.4}MnO_3$  (LSMO).

In this study, we report on the band diagrams of interfacial termination-layer controlled a half-metallic ferromagnetic oxide LSMO/Nb doped SrTiO<sub>3</sub> (Nb:STO) heterojunctions determined by *in situ* photoemission (PES) studies.

#### **Experimental**

LSMO/TiO2-Nb:STO and LSMO/SrO/Nb:STO (n-type and p-type LSMO/Nb:STO, respectively) heterojunctions were fabricated in a laser molecular beam epitaxy synchrotron-radiation chamber connected to а photoemission system at BL-2C. The Nb:STO substrate was annealed at 1050 °C and an oxygen pressure of  $1 \times$  $10^{-7}$ Torr to ensure an atomically flat TiO<sub>2</sub> layer terminated surface. For p-type structure, we initially deposited SrO on the TiO2-terminated Nb:STO substrate to change its termination from the TiO<sub>2</sub> to SrO layer. During LSMO depositions, the substrate temperatures and the ambient oxygen pressures were 1000 °C and  $1 \times 10^{-4}$ Torr, respectively. The film thicknesses were controlled on an atomic scale by monitoring the intensity oscillations of the reflection high-energy electron diffraction specular spot during growth. The PES spectra were taken in situ a total energy resolution of 150 meV in the energy range of 600 to 800 eV. The work functions  $(\phi_m)$  and electron affinities  $(\chi_i)$  were determined from the secondary electron emission spectra recorded with the He I (21.2 eV) resonance line.

# **Results and discussion**

The measurement of core-level spectra enables us to determine  $\Phi_{\rm B}$  formed at the heterojunctions directly. For both junctions, a peak shift towards a lower binding energy was clearly observed as the overlayer film thickness increased. Judging from the saturation level of the peak shift,  $\Phi_{B}$  could be estimated to be 1.2  $\pm$  0.1 and  $0.6 \pm 0.1$  eV for the *n*-type and *p*-type heterojunctions, respectively. The band diagrams for the (a) *n*-type and (b) p-type LSMO/Nb:STO heterojunctions derived from the present PES experiments, are illustrated in Fig. 1.  $\Phi_B$  of *n*-type LSMO/Nb:STO is much *higher* than the prediction from the Schottky-Mott rule  $(\phi_m - \chi_i)$  by 0.5 eV, indicating the formation of an "interface dipole" [2]. In contrast, for *p*-type LSMO/Nb:STO,  $\Phi_{\rm B}$  is *lower* by 0.4 eV. These results suggest that the direction of the interface dipole is inverted by changing the termination layer owing to the inversion of the polarity discontinuity at the polar/nonpolar interface.



FIG. 1. : Band diagrams derived from the present *in situ* PES measurements for (a) *n*-type and (b) *p*-type LSMO/Nb:STO Schottky junctions.

## **References**

N. Nakagawa *et al.*, Nature Mater. **5**, 204 (2006).
M. Minohara *et al.*, Appl. Phys. Lett. **90**, 132123 (2007).

\* kumigashira@sr.t.u-tokyo.ac.jp