

## Local structures of cubic $\text{EuTa}(\text{O},\text{N})_3$ oxynitride substituents

Shinichi KIKKAWA\*<sup>1</sup>, Teruki MOTOHASHI<sup>1</sup>, Yuji MASUBUCHI<sup>1</sup>,  
Takashi TAKEDA<sup>2</sup>, Kei-ichiro MURAI<sup>3</sup>, Akira YOSHIASA<sup>4</sup>

<sup>1</sup>Hokkaido Univ., Sapporo 060-8628, Japan

<sup>2</sup>NIMS, Ibaraki 305-0044, Japan

<sup>3</sup>Tokushima Univ., Tokushima 770-8506, Japan

<sup>4</sup>Kumamoto Univ., Kumamoto 860-8555, Japan

### Introduction

Tantalum oxynitride perovskites  $\text{ATaO}_2\text{N}$  (A: alkali-earth and rare-earth elements) have attracted much attention due to their unusual dielectric properties [1].  $\text{BaTaO}_2\text{N}$  and  $\text{SrTaO}_2\text{N}$  exhibit bulk permittivity ( $\kappa$ ) as large as 3000 ~ 5000. The dielectric properties seem to be related to the crystal structure, as  $\kappa$  of orthorhombic  $\text{CaTaO}_2\text{N}$  is much smaller ( $\approx 30$ ) than that of the Ba-analogue with cubic symmetry. For  $\text{SrTaO}_2\text{N}$ , on the other hand, the structure was refined based on either tetragonal [1,2] or cubic model [3]. It has not been well understood what material parameters are predominant for determining the crystal structure of tantalum oxynitride perovskites. To address this issue, we focused on  $\text{EuTa}(\text{O},\text{N})_3$ . This compound can be a structural reference for  $\text{SrTaO}_2\text{N}$ , since the ionic radius of  $\text{Eu}^{2+}$  is close to that of  $\text{Sr}^{2+}$ . In the present study, the structure and anionic composition of  $\text{EuTa}(\text{O},\text{N})_3$  were investigated. A series of the  $\text{Ca}_{1-x}\text{Eu}_x\text{Ta}(\text{O},\text{N})_3$  system was synthesized and characterized in order to clarify how the properties are influenced upon substituting Eu for Ca.

### Experimental

$\text{Ca}_{1-x}\text{Eu}_x\text{Ta}(\text{O},\text{N})_3$  samples with  $x = 0, 0.2, 0.4, 0.6, 0.8,$  and  $1.0$  were synthesized by ammonia nitridation at  $1000^\circ\text{C}$  via a citrate precursor route, as described elsewhere [4]. Nitrided products were characterized by X-ray powder diffraction (XRD). The anionic composition was determined using an oxygen/nitrogen analyzer. X-ray absorption of Eu  $L_{\text{III}}$ - and Ta  $L_{\text{III}}$ -edges was measured in transmission mode at the beam line 9C in Photon Factory.

### Results and discussion

XRD analysis indicated that all of the  $\text{Ca}_{1-x}\text{Eu}_x\text{Ta}(\text{O},\text{N})_3$  products are of single phase. The products with  $x < 0.4$  were apparently orthorhombic, while those with  $x \geq 0.4$  crystallized in a cubic structure. The anionic composition was found to be essentially  $\text{O}_2\text{N}$  and independent of Eu content ( $x$ ). Figure 1 shows Eu  $L_{\text{III}}$ -edge XANES spectra of  $\text{Ca}_{1-x}\text{Eu}_x\text{Ta}(\text{O},\text{N})_3$ . The spectra consist of two peaks in all the Eu-containing products. Since the low-energy ( $E \approx 6970$  eV) and high-energy ( $E \approx 6980$  eV) peaks are similar to those for the  $\text{EuCl}_2$  and  $\text{Eu}_2\text{O}_3$  standards, the former and the latter are accordingly attributed to the  $2p_{3/2} \rightarrow 5d$  transition of  $\text{Eu}^{2+}$  and  $\text{Eu}^{3+}$ , respectively. Therefore, europium ions are in mixed valent between +2 and +3.

The  $\text{Eu}^{3+}$  peak is large in  $x = 0.2$  because its ionic size is small enough to occupy the  $\text{Ca}^{2+}$  site, while its intensity systematically decreases with increasing  $x$ . In  $x = 1.0$  ( $\text{EuTaO}_2\text{N}$ ), europium is mainly divalent with a small amount of trivalent species.

The  $\text{Eu}^{2+}/\text{Eu}^{3+}$  coexistence is easily explained if the Eu-containing products are assumed to be slightly off-stoichiometric with N-rich compositions. The presence of excess nitride anions in  $\text{EuTaO}_2\text{N}$  was indeed reported previously [5]. Another possible explanation is that tantalum is also mixed-valent between  $\text{Ta}^{4+}$  and  $\text{Ta}^{5+}$  so as to compensate extra positive charges of europium. It was not possible to determine the Ta valence by means of XANES spectroscopy without ambiguity, because the positions of  $L_{\text{III}}$ -edge are close to each other for  $\text{Ta}^{4+}$  and  $\text{Ta}^{5+}$ .

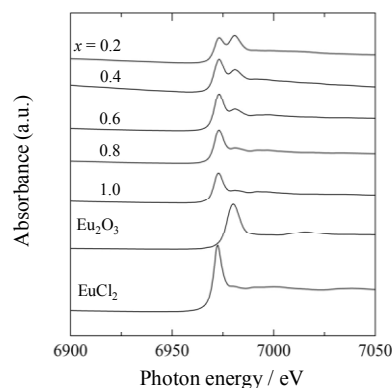


Fig. 1 Eu  $L_{\text{III}}$ -edge XANES spectra of  $\text{Ca}_{1-x}\text{Eu}_x\text{Ta}(\text{O},\text{N})_3$  with  $x = 0.2, 0.4, 0.6, 0.8,$  and  $1.0$ . The reference spectra of  $\text{Eu}_2\text{O}_3$  ( $\text{Eu}^{3+}$ ) and  $\text{EuCl}_2$  ( $\text{Eu}^{2+}$ ) are also shown.

### References

- [1] Y. -I. Kim *et al.*, Chem. Mater. 16, 1267 (2004).
- [2] R. Marchand *et al.*, Rev. Int. Hautes Temp. Refractaires 23, 11 (1986).
- [3] F. Pors *et al.*, Rev. Int. Hautes Temp. Refractaires 24, 239 (1987).
- [4] Y. Hamade *et al.*, Mater. Res. Bull. (2009) submitted.
- [5] A. B. Jorge *et al.*, J. Am. Soc. Chem. 130, 12572 (2008).

\* kikkawa@eng.hokudai.ac.jp