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Study on the Crystal and Electronic Structures of the Layered Li₂MO₃-LiMO₂ Materials in Li de-intercalation process

<u>Hironori Kobayashi</u>^{1,2}, Masahiro Shikano¹, Hiroyuki Kageyama¹, Yuki Takenaka², Yoshinori Arachi², and Hiroaki Nitani³

¹ Research Institute for Ubiquitous Energy Devices, AIST, Ikeda, Osaka, 563-8577 Japan

² Department of Chemistry and Materials Engineering, Faculty of Chemistry, Materials and

Bioengineering, Kansai University, Suita, Osaka, 564-8680 Japan

³Institute of Materials Structure Science, KEK, Tsukuba, 305-0801 Japan

1 Introduction

The layered Li₂MO₃-LiMO₂ materials are one of the promising positive electrode materials of lithium secondary battery because of their large capacity when operated above 4.6 V [1]. In this system, especially, Li[Ni_{0.17}Li_{0.2}Co_{0.07}Mn_{0.56}]O₂ displays a initial discharge capacity of c.a. 280 mAh/g in the voltage range of 2.5 to 4.8 V and keep a reversible capacity of c.a. 250 mAh/g after 50 cycles [2]. Several papers have reported on the mechanism why these materials show large reversible capacity. However, the initial charge and discharge process are still ambiguous. We have reported on the characteristic structural change during Li de-intercalation for $LiNi_{1/2}Mn_{1/2}O_2$ [3-4] using synchrotron radiation. Detailed information on the crystal and electronic structures above 4.6 V is very important in order to improve the calendar life and thermal stability of these materials and, therefore, the structural and electronic changes of $Li_{1.20}Ni_{0.17}Co_{0.10}Mn_{0.53}O_2$ (sample A) and $Li_{1,01}Ni_{0,49}Co_{0,21}Mn_{0,29}O_2$ (sample B) in Li deintercalation process were studied in this paper.

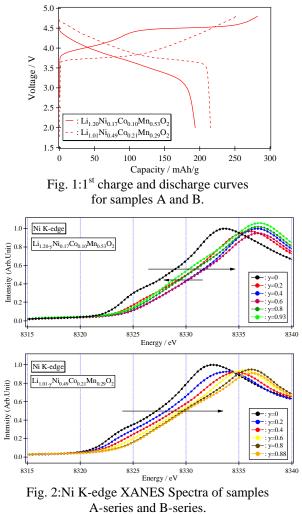
2 Experiment

The de-lithiated Li_{1,20-y}Ni_{0,17}Co_{0,10}Mn_{0,53}O₂ (samples A-series) and Li_{1,01-y}Ni_{0,49}Co_{0,21}Mn_{0,29}O₂ (samples B-series) were electrochemically prepared using coin-type cells with Li/1M LiPF₆ in EC:DMC(1:2)/samples. Crystal and electronic structures were investigated by synchrotron XAFS (BL7C at PF) measurements. The bondlength were determined using the analysis programs REX2000.

3 Results and Discussion

Figure 1 shows the 1^{st} charge and discharge curves of samples A and B in the voltage range of 2.0 and 4.8 V. The Li/sample A cell showed the 1^{st} charge capacity of 282 mAh/g and gave the characteristic plateau region around 4.5 V. Figure 2 shows the Ni K-edge XANES Spectra of samples A-series and B-series with Li de-intercalation. For samples A-series, Ni K-edge XANES spectra shift to higher energy up to y=0.4-0.6 and then shift to lower energy up to y=0.93. This means that the valence state of Ni increased to 4+ and then decreased with Li de-intercalation. For samples B-series, Ni K-edge XANES spectra display continuous shifts to y=0.88. This

means that the valence state of Ni increased to 4+ with Li de-intercalation. These results indicated that Li de-intercalated mechanism is different between samples A and B.



References

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